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# Tugging at Polymers in Turbulent Flow

## *The Polymers Tug Back*

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# Drag Reduction

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Experimental facts:

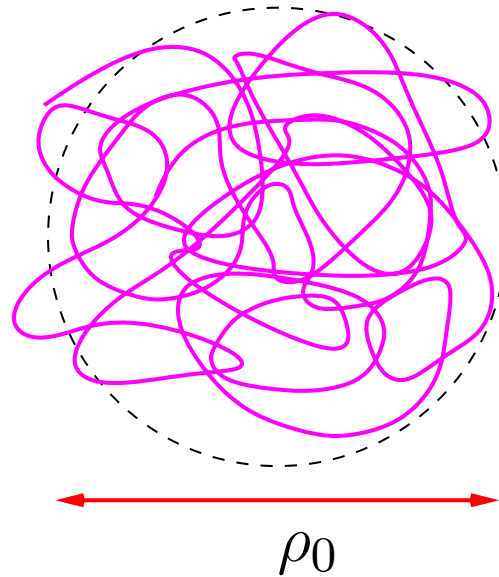
- **Toms** (1948) observed that the addition of  $\sim 10$  ppm polymers to **turbulent pipe flow** reduced the pressure drop substantially.
- Typical example: polyethylene oxide in water: **18 ppm** (by weight) reduces drag by **33% !!**
- Qualitative understanding: classic review of **Lumley** (1969) uses dimensional analysis (**polymer size, viscosity, etc.**) to predict magnitude and onset of effect.
- **Molecular scales matter!** Mystery?
- Stretched state: **Einstein's effective viscosity** [**Hinch** (1977)].

# Polymers

Polymers are long chains of molecules. Random walk at rest.

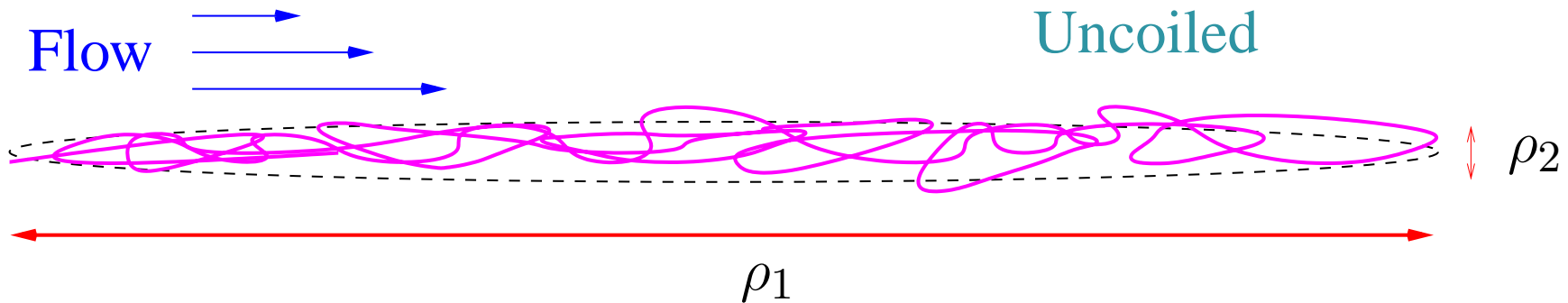
No flow

Coiled



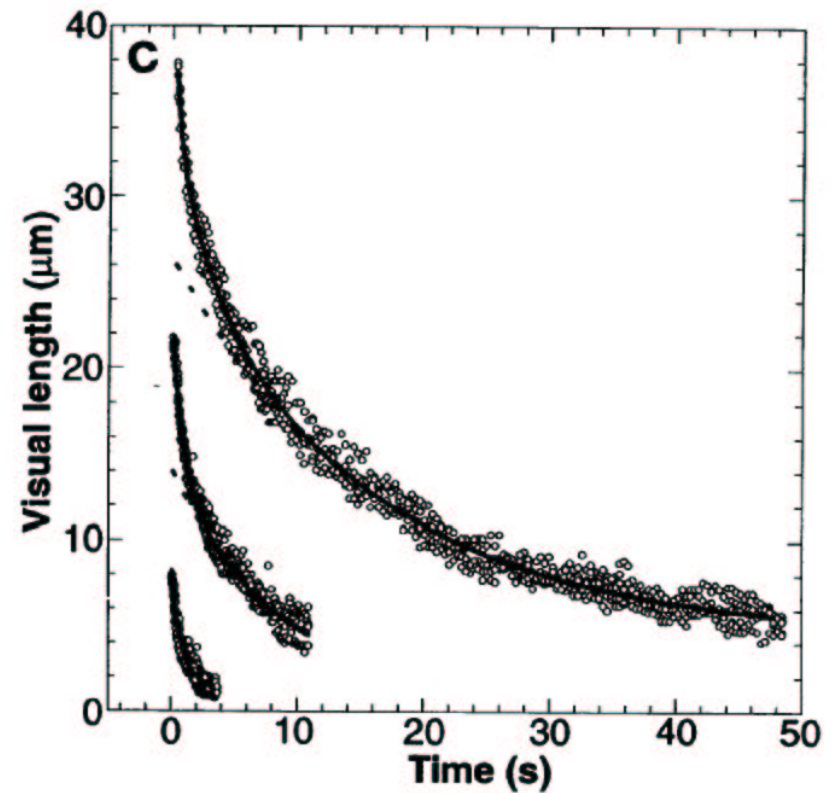
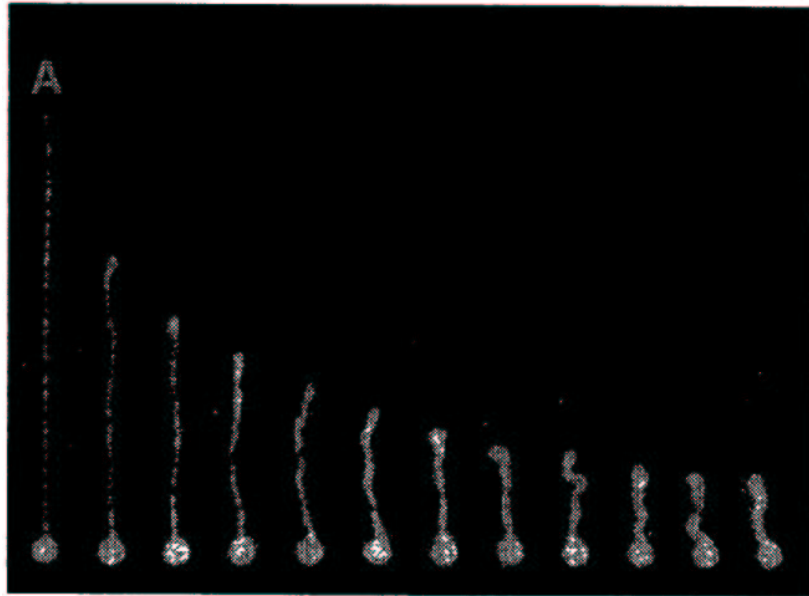
Flow

Uncoiled



# Relaxation of Polymers

Pretty close to exponential...



From Perkins *et al.*, *Nature* (1994)

# Constitutive Models

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Model polymers in dilute solution as a continuum:

Stress on the fluid.

How does the stress tensor  $\mathbf{T}$  depend on the state of the fluid?

- Elastic:

$$T_{ij} = \eta \gamma_{ij}, \quad \gamma \equiv \text{strain (deformation) tensor.}$$

- Viscous or Newtonian:

$$T_{ij} = \mu \dot{\gamma}_{ij}, \quad \dot{\gamma} \equiv \nabla \mathbf{u} + (\nabla \mathbf{u})^T$$

- Viscoelastic:

$$T_{ij} = \int_{-\infty}^t G(t - t') \dot{\gamma}_{ij}(t') dt'$$

# Maxwell Model

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Exponentially-decaying memory:

$$T_{ij} = (\mu/\tau) \int_{-\infty}^t e^{-(t-t')/\tau} \dot{\gamma}_{ij}(t') dt'$$

Reformulate as differential equation for  $\mathbb{T}$ :

$$\tau \dot{T}_{ij} = \mu \dot{\gamma}_{ij} - T_{ij}$$

Problem: **not frame-indifferent!**

Not good as a fluid relation. Remedied by introducing a frame-independent (**Oldroyd**) derivative

$$\dot{\mathbb{T}} \implies \mathcal{D}\mathbb{T} \equiv \frac{\partial \mathbb{T}}{\partial t} + \mathbf{u} \cdot \nabla \mathbb{T} - (\mathbb{T} \cdot \nabla \mathbf{u} + (\nabla \mathbf{u})^T \cdot \mathbb{T})$$

# The Equations of Motion

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Couple stress to Navier–Stokes for an incompressible fluid:

$$\frac{\partial \mathbf{u}}{\partial t} + \mathbf{u} \cdot \nabla \mathbf{u} + \nabla p = \nu \nabla^2 \mathbf{u} + \frac{s}{\tau} \nabla \cdot \mathbb{A};$$
$$\mathcal{D}\mathbb{A} = -\frac{2}{\tau} (\mathbb{A} - \rho_0^2 \mathbb{I}); \quad \nabla \cdot \mathbf{u} = 0,$$

where  $\mathbb{A}$  is equal to  $\mathbb{T}$  up to constants, and can be regarded as the local **deformation** of the polymers, with  $\mathbb{A} = \rho_0^2 \mathbb{I}$  at rest.

Can be derived from a kinetic model of “**Hookean dumbbells**.”

More generally: allow **nonlinear saturation** of the length of polymers (**FENE**-type models)

$$\mathcal{D}\mathbb{A} = -\frac{1}{\tau} (f(A) \mathbb{A} - \rho_0^2 \mathbb{I}).$$

# Evolution of the Principal Axes

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The polymer **conformation tensor**  $\mathbb{A}$  can be diagonalized, with orthonormal eigenvectors  $\mathbf{e}_\alpha$  and eigenvalues  $(\rho_\alpha)^2$  that evolve according to

$$\frac{d\rho_\alpha}{dt} = \lambda_\alpha \rho_\alpha - \frac{1}{\tau} (f(\|\rho\|^2)\rho_\alpha - \rho_0^2/\rho_\alpha),$$

$$\lambda_\alpha(t, \mathbf{x}) \equiv \mathbf{e}_\alpha \cdot \nabla \mathbf{u} \cdot \mathbf{e}_\alpha, \quad d/dt = \partial/\partial t + \mathbf{u} \cdot \nabla.$$

The  $\rho_\alpha$  are the **lengths of the principal axes** of the ellipsoid delineating the deformation of the polymer.

If the flow is **smooth**, the polymers tend to align with the dominant stretching direction, so we consider only the major axis:

$$\boxed{\frac{d\rho}{dt} = \lambda \rho - \frac{1}{\tau} (f(\rho^2)\rho - \rho_0^2/\rho)}$$



# Turbulence! (well...sort of)

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For turbulence, a locally smooth regime is achieved when the **viscous scale** is much longer than **the polymer length**.

In that case, model the velocity field as a **Gaussian random variable** representing a smooth straining field  $\lambda(t)$  that changes rapidly;  $\lambda$  satisfies

$$\langle \lambda(t)\lambda(t') \rangle - \bar{\lambda}^2 = \delta(t - t') \Delta; \quad \langle \lambda(t) \rangle = \bar{\lambda},$$

where the angle brackets denote an average over  $\lambda$ .

The variable  $\lambda(t)$  is  **$\delta$ -correlated** in time, which means that it forgets about its previous state immediately. It has mean  $\bar{\lambda}$  and standard deviation  $\Delta$ .

This “slightly” artificial situation has great analytical advantages.

# Fokker–Planck Equation

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Because the distribution of  $\lambda(t)$  is Gaussian and  $\delta$ -correlated, can obtain a **Fokker–Planck equation** for the distribution  $\mathcal{P}(t, \rho)$  of the major axis:

$$\mathcal{Z}(t; \mu) = \langle \exp(i \mu \rho) \rangle$$

We can then derive an equation of motion for  $\mathcal{Z}$  and average.

**Gaussian integration by parts** allows evaluation of terms of the form  $\langle \lambda \mathcal{Z} \rangle$ .

Inverse Fourier transformation of  $\mathcal{Z}$  with respect to  $\mu$  then gives the equation of motion for  $\mathcal{P}(t, \rho)$ , **the PDF of  $\rho$** , (Chertkov, 2000)

$$\partial_t \mathcal{P} = \frac{1}{2} \Delta \partial_\rho \rho \partial_\rho \mathcal{P} - \bar{\lambda} \partial_\rho \rho \mathcal{P} + \frac{1}{\tau} \partial_\rho (f(\rho^2) \rho - \rho_0^2 / \rho) \mathcal{P}$$

# Stationary Distribution: Coiled State

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Natural thing to do is to look for **stationary solutions** of the F–P equation. Assume first that the polymers are uncoiled. Neglect nonlinear relaxation:  $f = 1$  (Hookean springs).

$$\mathcal{P}_c(\rho) \sim \rho^{-1-2(\xi-\zeta)} \exp(-\xi \rho_0^2 / \rho^2)$$

where  $\rho$  is normalized by  $\rho_0$ , and

$$\xi \equiv 1/\Delta\tau, \quad \zeta \equiv \bar{\lambda}/\Delta$$

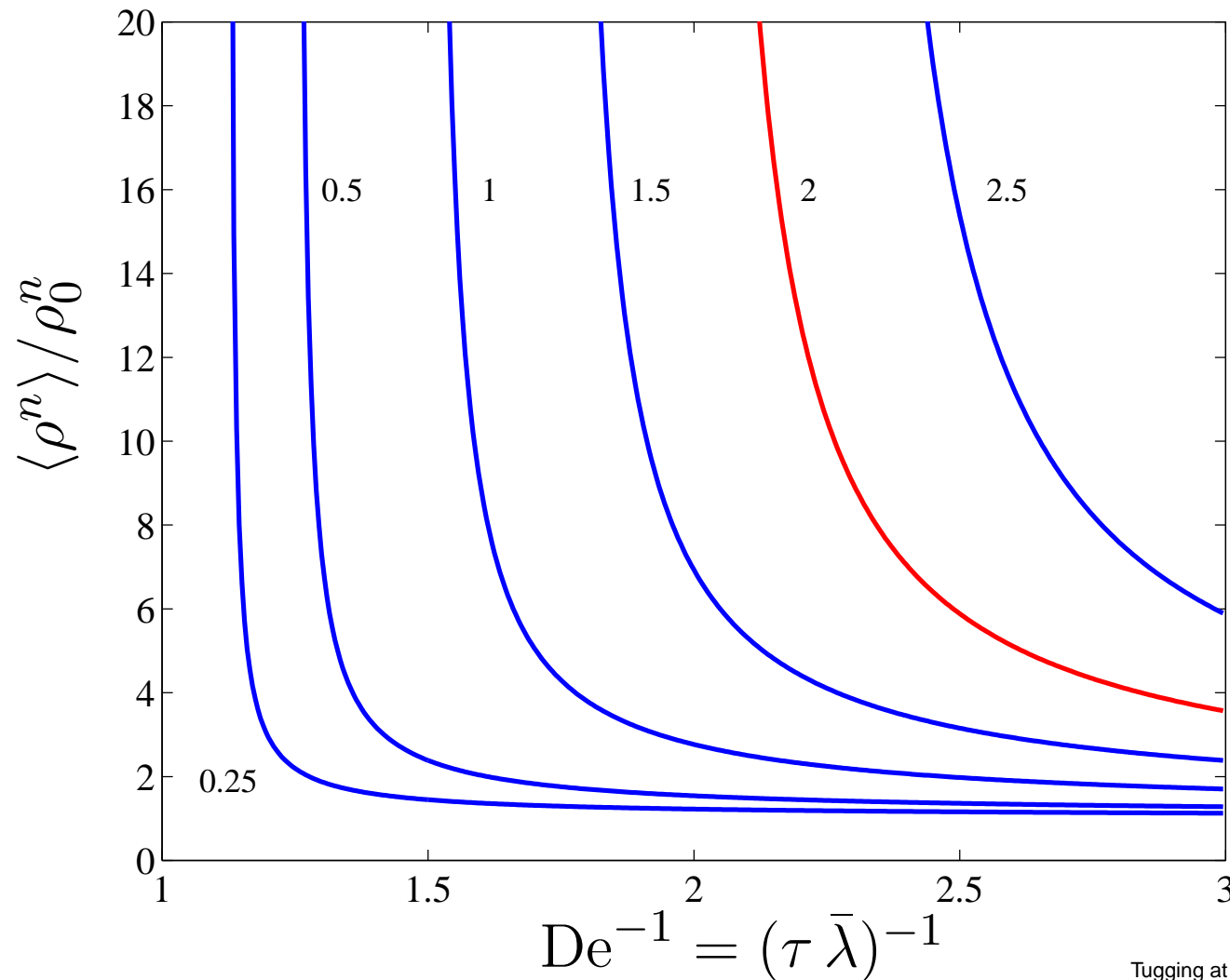
are dimensionless numbers.

$\zeta/\xi = \text{De}$ , the **Deborah number**, the ratio of the **polymer relaxation timescale**,  $\tau$ , over the **advection timescale**,  $1/\bar{\lambda}$ .

Large  $\text{De} \implies$  polymers more affected.

# Moments of the Distribution

Below the coil-stretch transition at  $De = 1$  (with  $\bar{\lambda}/\Delta = 1$ ).



# Stationary Distribution: Stretched State

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Assuming a **FENE**-type (**F**inite **E**xtension **N**onlinear **E**lastic) model which limits the length of the polymers to  $\rho_m$ ,

$$f(\rho^2) = \frac{\rho_m^2 - \rho_0^2}{\rho_m^2 - \rho^2}.$$

Can find equilibrium distribution

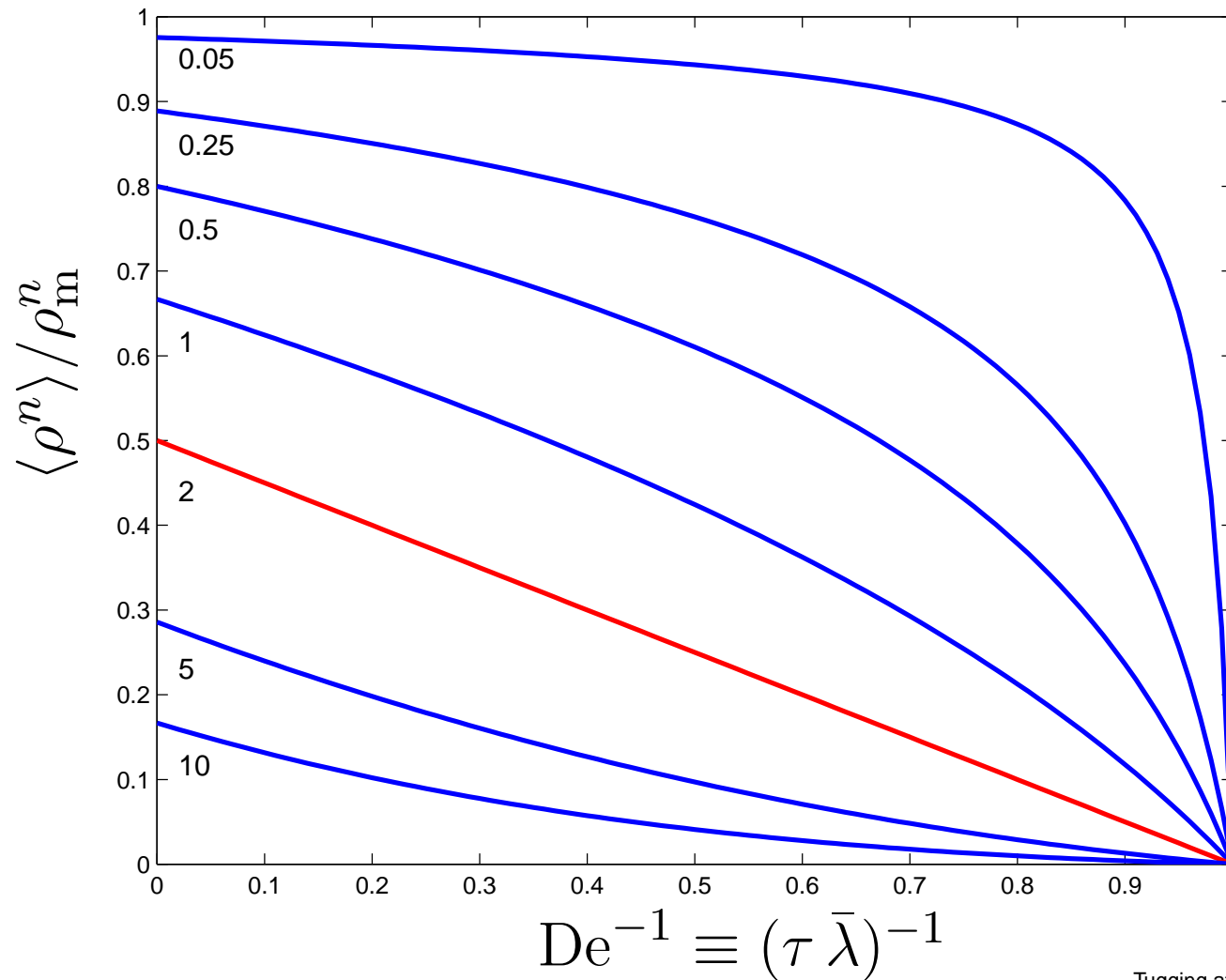
$$\mathcal{P}_s(\rho) \sim \rho^{-1+2(\bar{\lambda}/\Delta-\xi)} (1 - \rho^2)^\xi$$

where  $\rho$  is normalized by  $\rho_m$ , and  $\xi \equiv 1/\tau\Delta$ .

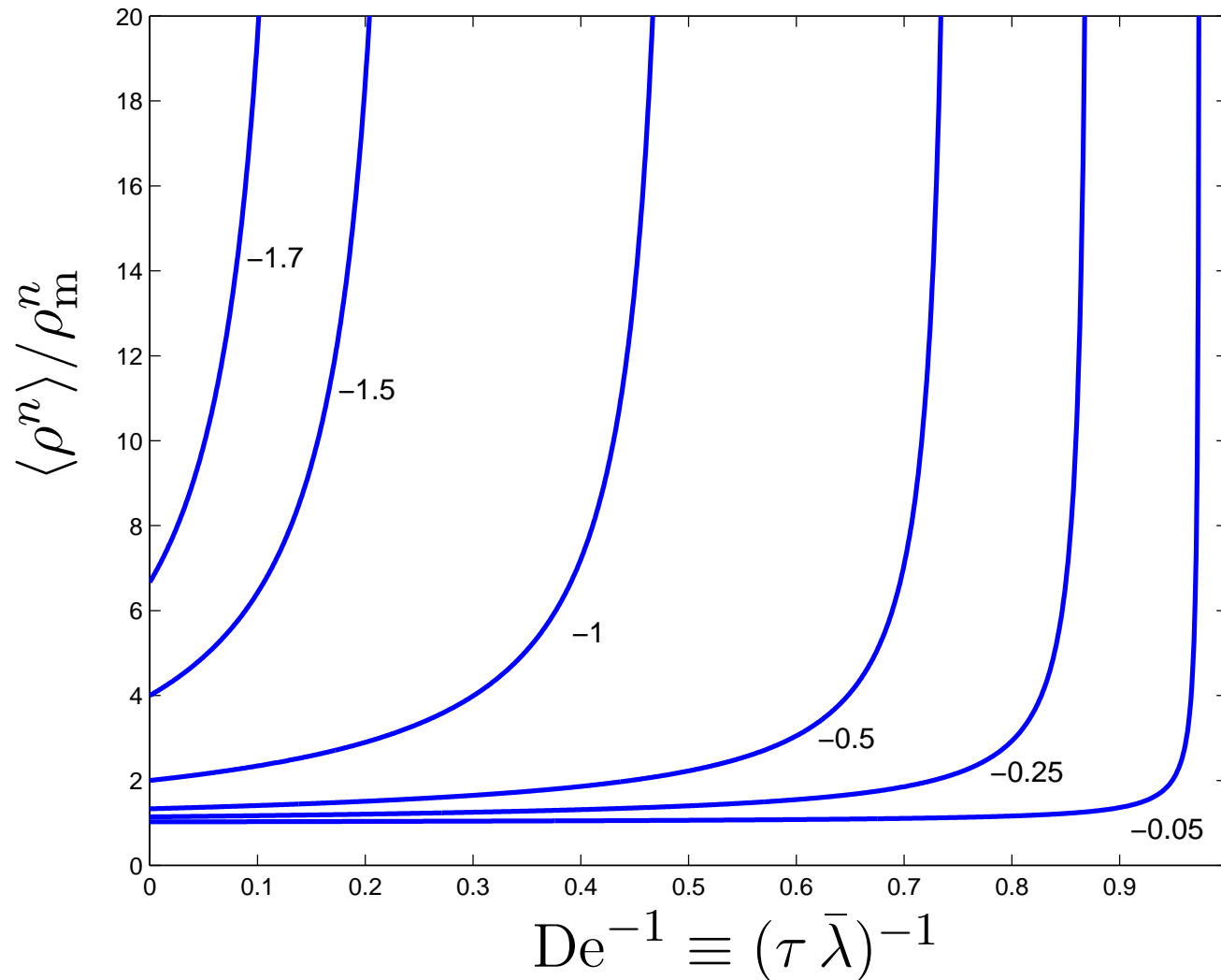
$\implies$  Cutoff in PDF for  $\rho > \rho_m$ .

# Moments of the Distribution

Coil-stretch transition at  $De = 1$  (with  $\bar{\lambda}/\Delta = 1$ ).



# Negative moments



# Breaking Point

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The PDF of length can also be used to estimate the **fraction of polymers that break**.

Assume a polymer breaks if the tension  $F(\rho^2) = f(\rho^2)\rho/\tau$  exceeds a critical value,  $F_c$ .

To that critical tension corresponds a critical length,  $\rho_c$ , obtained by solving

$$F_c = \frac{1}{\tau} f(\rho_c^2)\rho_c = \frac{1}{\tau} \frac{\rho_m^2}{\rho_m^2 - \rho_c^2} \rho_c,$$
$$\implies \rho_c = \frac{\rho_m^2}{2\tau F_c} \left( \sqrt{1 + (2\tau F_c/\rho_m)^2} - 1 \right).$$



# The Survivors

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The fraction of polymers that **survive** is equal to the **fraction shorter than  $\rho_c$** , obtained by integrating the PDF from 0 to  $\rho_c$ ,

$$\text{Prob}(\rho < \rho_c) = \frac{B(\rho_c^2/\rho_m^2; \zeta - \xi, \xi + 1)}{B(\zeta - \xi, \xi + 1)},$$

where

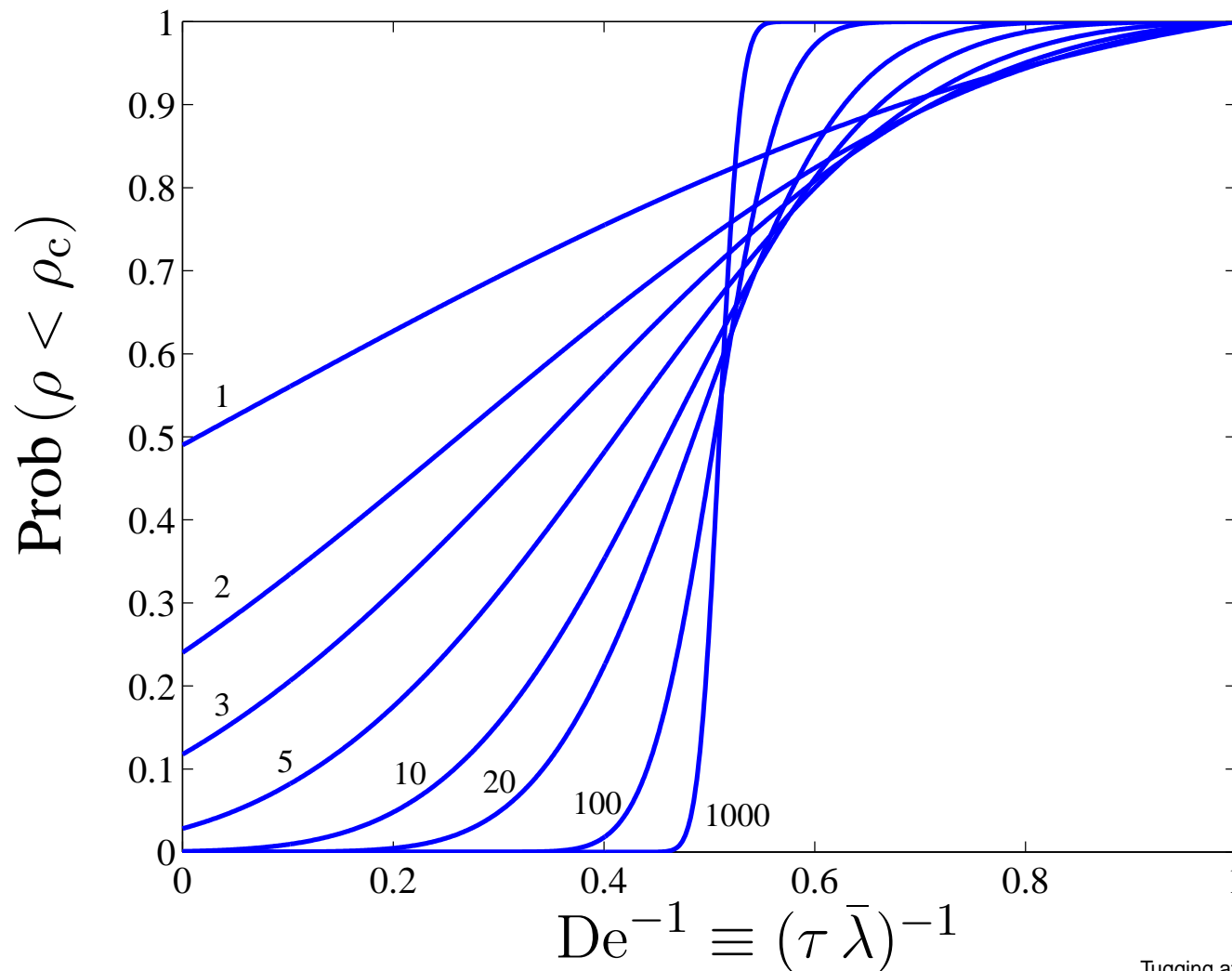
$$B(a, b) \equiv \Gamma(a)\Gamma(b)/\Gamma(a + b)$$

is the **beta function**, and  $B(z; a, b)$  is the incomplete beta function.

Since  $B(1; a, b) = B(a, b)$ , for  $\rho_c = \rho_m$  none of the polymers break.

# Surviving Fraction as a function of $De^{-1}$

For different values of  $\zeta \equiv \bar{\lambda}/\Delta$  with  $\rho_c/\rho_m = 0.7$ .



# The Survivors: Strong Turbulence

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For **large Deborah number**,

$$\text{Prob}(\rho < \rho_c) \simeq (\rho_c/\rho_m)^{2\zeta},$$

which is not small if  $\zeta$  is not too large.

$\implies$  Some polymers **survive breakage** for large  $\bar{\lambda}\tau$  if the **fluctuations** in  $\bar{\lambda}$  (given by  $\Delta$ ) are also large.

Of course, this treatment is for **one correlation time** of the turbulence. In reality a given polymer is exposed to **many different random strains**.

Survival prob. decays roughly as  $(\rho_c/\rho_m)^{2\zeta N}$ , where  $N = t/t_{\text{corr}}$  with  $t_{\text{corr}}$  the correlation time.

# Ongoing Research

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- Model polymer as flexible chains instead of rods.
- Self-consistency: backreaction of the polymers on the fluid.
- Incorporate Einstein model of effective viscosity.
- Non-Gaussian statistics: Path integral formalism.
- Compressibility.
- Magnetic dynamo.
- Statistics of curvature.